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## PROTONIC CONDUCTION AND O-H BONDING STATE IN PHOSPHATE GLASSES

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**Abstract:** Protons in alkaline-earth phosphate glasses give much higher mobility than those in silicate glasses. This is caused by the difference in O-H bonding state. Protons in P-OH form a strong hydrogen bonding with oxygen anion, resulting in formation of weak O-H bond.

The electrical conductivity of the glasses is proportional to the square of the proton concentration. The activation energy for the conduction is related to  $\nu_{OH}$ , peak wavenumber of infrared absorption band (around  $\sim 3000\text{ cm}^{-1}$ ), due to fundamental O-H stretching vibration. The mobility of protons increases with decreasing  $\nu_{OH}$ , which depends on the species of ions included in the phosphate glasses.

Protonic conduction in glasses has attracted much attention as the search for developing fast proton conductive glasses which are applicable to a solid electrolyte for  $H_2$ - $O_2$  fuel cell and for a  $H_2$  gas sensor. In most oxide glasses, electric charge carriers are mobile alkali ions or electrons, but not protons.

Glasses, in most cases, contain a small amount of protons as hydroxyl groups(-OH), which was first studied extensively by Scholze<sup>1</sup> for silicate glasses. Some phosphate glasses exhibit characteristic unique behaviors in the crystallization<sup>2</sup>. The protonic conduction is one of the characteristic properties of phosphate glasses. Since the O-H bonding in phosphate glasses is generally weak owing to the formation of hydrogen bonding, protons in this case are mobile.

### Experimental

Glasses were prepared by melting a given mixture of reagents, such as  $H_3PO_4$  and alkaline-earth carbonates, in a platinum crucible at 1200 to 1300°C for various times. The proton concentration in the glasses was controlled by the melting duration times<sup>3</sup>. Direct current conductivity measurements were made with a vibrating reed electrometer (impedance  $10^{16}$ ) over a temperature range from 80°C to 200°C.

Proton content in the glasses was determined by infrared spectroscopy, where the molar absorption coefficient is assumed to depend on  $\xi_{H_2O}$  as shown in Fig.1.

### Results and Discussion

Protons in glass form -OH groups. Fig.2 shows an example of IR spectrum of glass plate specimen for  $Ca(PO_3)_2$  glasses. Scholze<sup>1</sup> established that there are three main absorption bands due to OH or "water" in glass, i.e., Band 1 in the range  $3640-3390\text{ cm}^{-1}$ , Band 2 ( $3000-2600\text{ cm}^{-1}$ ), and Band 3 ( $2350\text{ cm}^{-1}$ ). However, Band 3 was excluded from the present discussion because the existence of Band 3 is not certain yet and various absorption bands due to network formers (i.e., P-O-P etc) are overlapped. Silica glass has only Band 1, but  $Na_2O-CaO-SiO_2$  glasses have all three bands. In alkaline-earth phosphate glasses, Band 1 exists but is not clearly observed owing to the overlapping with Band 2; the deconvolution result shows clearly the existence of Band 1, although the Band 1 is smaller than the Band 2. Band 1 is due to hydrogen bonding-free OH and Band 2 is to hydrogen bonded OH. It was found that protons in Band 2 are much more mobile than those in Band 1. The electrical conductivity in alkaline-earth phosphate glasses was found to increase with the square of proton concentration in Band 2 as expressed by equation (1)<sup>4</sup>,

$$\sigma_{417} = A_0 [H^*]^2 \quad \text{-----} \quad (1)$$

where  $\sigma_{417}$  is the conductivity(S/cm) at 417K( $10^3/T=2.4$ ),  $[H^*]$  is proton concentration(mol/l), and  $A_0$  is a constant which is a measure of mobility. The value of  $A_0$  was found to vary to a great extent with  $\nu_{OH}$  as shown in Fig.3.

$$\log A_0 = -0.00937 \nu_{OH} + 17.1 \quad \text{-----} \quad (2)$$

The experimental activation energy for the electrical conduction E was assumed to be expressed by equation (3),

$$E = E_0 + E_1 \quad \text{-----} \quad (3)$$

where,  $E_0$  is an activation energy at a unit concentration of the proton ( $[H^*]=1$ ) and  $E_1$  depends on the proton concentration.

It is found that  $E_0$  increases linealy with increasing  $\nu_{OH}$  and  $E_1$  is propotional to  $\log[H^*]$ . The major portion of E is determined by  $E_0$  ( $E_0 \gg E_1$ ); E is estimated to be decreased to 9/10 times when  $[H^*]$  increases by ten times the initial concentration. Thus, E is mainly determined by  $\nu_{OH}$ . The O-H bonding strength decreases with increasing hydrogen bonding, i.e.,with decreasing  $\nu_{OH}$ . Generally, in phosphate glasses proton concentration is considerably higher and hydrogen bonding is stronger than in silicate glasses. This is the reason why phosphate glasses exhibit protonic conduction.

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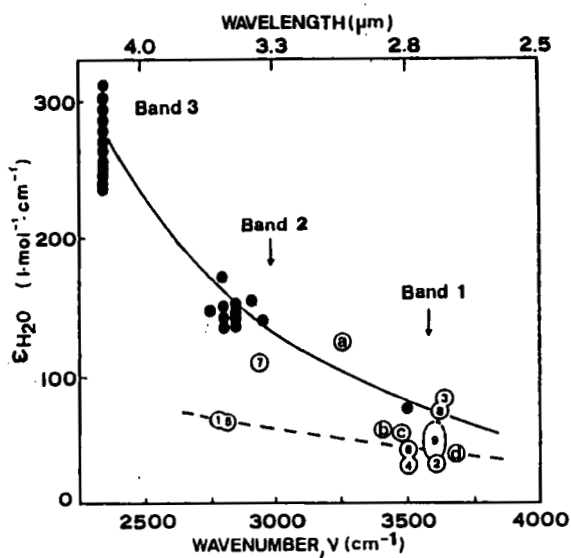


Fig.1, Relation between  $\epsilon_{H_2O}$  and  $\nu_{OH}$ .

(plotted from the data in references so far obtained. The solid line should be used for the separate determination of the Bands.)

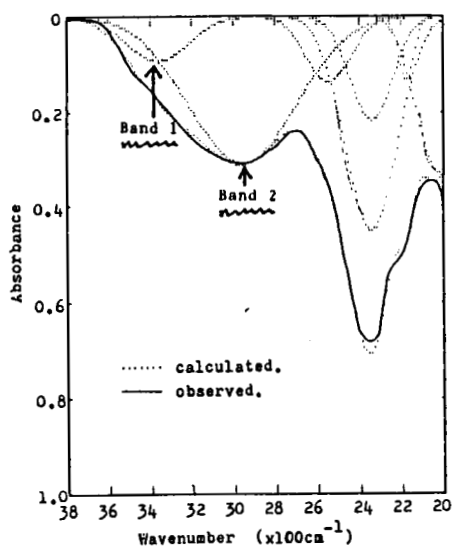


Fig.2, Example of deconvolution of IR spectrum for  $Ca(PO_3)_2$  glass.

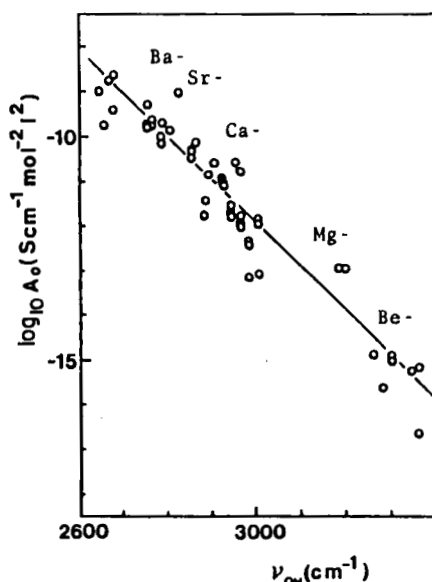


Fig.3,  $\log A_0$  vs  $\nu_{OH}$  plot  
The position of respective alkaline-earth metaphosphate glasses are indicated.